Determination of pore-size distribution in low-dielectric thin films

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Positronium annihilation lifetime spectroscopy is used to determine the pore-size distribution in low-dielectric thin films of mesoporous methylsilsesquioxane. A physical model of positronium trapping and annihilating in isolated pores is presented. The systematic dependence of the deduced pore-size distribution on pore shape/dimensionality and sample temperature is predicted using a simple quantum mechanical calculation of positronium annihilation in a rectangular pore. A comparison with an electron microscope image is presented. © 2000 American Institute of Physics.

There is currently a great deal of interest in introducing and characterizing nanometer-sized voids into thin silica and polymer films. Such porous films are being intensely pursued by the microelectronics industry as a strategy for reducing the dielectric constant of interlayer insulators in microelectronic devices. Unfortunately, there are relatively few techniques capable of probing the average pore size and/or the pore size distribution in submicron films on thick substrates. This is particularly true if the voids are closed (not interconnected) so that gas absorption techniques are not available. Neutron scattering1 and positronium annihilation lifetime spectroscopy (PALS)2 have recently been used to determine an average pore size in silica films and Doppler broadening positron spectroscopy3 has been used to probe open volume in silsesquioxane films. None of the methods can determine a pore-size distribution in porous silica (this is apparently a fundamental limitation in neutron scattering, whereas with PALS the limitation is a result of the complete interconnectness of the silica pores2). In closed-pore systems PALS may be uniquely capable of deducing a pore-size distribution, even in films buried under metal overlayers. The goal of this letter is to develop this PALS capability in a porous film that nominally has closed pores.

The sample films studied were formed by spin casting a homogeneous mixture of poly(methylsilsesquioxane) (MSSQ) and 10% by weight poly(caprolactone) (PCL), a degradable porogen. Upon heating to 250 °C the MSSQ prepolymer polymerizes and crosslinks, causing the PCL phase to separate into domains. At 430 °C the PCL has degraded and volatilized, leaving behind a mesoporous, 0.8-μm-thick MSSQ film4 with a dielectric constant of 2.5. We have performed a detailed PALS study of this system for varying PCL weight fractions, sample temperature, and processing conditions. Our focus here is to develop a physical model of the diffusion, trapping, and annihilation of positronium (Ps) in these films in order to determine the pore-size distribution from the PALS spectrum.

In using PALS with thin films5,6 a focused beam of several keV positrons forms positronium (Ps, the electron-positron bound state) over a distribution of depths that depends on the beam energy. Ps inherently localizes in the pores where its natural lifetime of 142 ns is reduced by annihilation with molecular electrons during collisions with the pore surfaces. The collisionally reduced lifetime is correlated with void size.2 This correlation is the key feature that permits a PALS lifetime distribution to be transformed into a pore-size distribution if Ps is trapped in isolated voids of varying sizes. This was not the case in the interconnected voids of porous silica films2 where the highly mobile Ps atoms all sample the same average pore size and hence annihilate with a single, average lifetime. Porous MSSQ presents a distribution of Ps lifetimes.

Lifetime spectra acquired before and after decomposition of the PCL are compared in Fig. 1. Data are typically acquired with an 8000 channel time-to-digital converter over a time range of 1.25 μs. Initially we used the standard program POSFIT to fit discrete lifetimes to this spectrum. Prior to PCL decomposition the “unfoamed” film presents lifetimes consistent with those in bulk MSSQ (0.4 ns, 1.5 ns, 6 ns, and a low intensity component at 140 ns consistent with back-scattered positrons forming Ps in vacuum), but no mesoporous lifetime components (in the range 10–120 ns) are observed. The sample foamed at 430 °C shows long-lived events that cannot be fitted with a single lifetime. At least two components (around 45 and 120 ns) are required in the POSFIT fitting, with the long lifetime component being indicative of diffusion and escape of Ps into the vacuum. This suspicion was confirmed by spectra acquired after the application of an 80 nm, sputter-deposited, Al capping layer that confines Ps to the porous film. The 120 ns component due to
escaping Ps disappeared as expected and lifetime components around 20 and 60 ns emerged, suggesting that only a fraction of the pores are interconnected. Thus, these films may have a complex, partially interconnected, pore structure. We will use the capped film spectra to eliminate specious results from the escape of Ps through the surface.

The lifetime histograms shown in Fig. 1 are given by \( \frac{dN_{Ps}}{dt} \), where \( N_{Ps} \) is the total number of Ps annihilation events recorded. We use the continuum fitting program, CONTIN, specialized for exponential lifetime analysis to obtain from these data a Ps lifetime distribution, \( \frac{dN_{Ps}}{dt} \), where \( t \) is the Ps lifetime (see Fig. 2). The peak widths in Fig. 2 are very sensitive to certain deconvolution parameters and should only be considered as typical of a class of statistically acceptable, bimodal lifetime distributions. The important result here is that both CONTIN and POSFIT find at least two, separable, lifetime components.

Given a statistically acceptable Ps lifetime distribution, \( \frac{dN_{Ps}}{dt} \), the goal is to transform this distribution into a distribution of void sizes. More precisely, we can extract a relative distribution of specific void volume (void volume per unit sample volume) as a function of the void’s mean-free path, \( l \), which is given classically by \( l = 4V_{void}/S_{void} \), where \( V_{void}/S_{void} \) is the volume-to-surface area ratio in a particular void. We introduce \( l \) as a linear measure of pore size that in the large-pore limit does not depend on any assumed model of pore shape. To illustrate this point we have calculated \( \tau \) vs \( l \) for two different rectangular pore model geometries: three-dimensional (3D) cubes, and two-dimensional (2D) infinitely long square channels. As can be seen in Fig. 3 the curves are slightly model dependent in the quantum mechanical regime but merge for large pores.

We first consider the case of a specific but arbitrary void geometry for which the Ps lifetime in the void at a given temperature is a unique function of \( l \). Given the function \( \tau(l) \) for this geometry we can determine the size distribution of pores in which Ps annihilates,

\[
\frac{dN_{Ps}}{dl} = \frac{dN_{Ps}}{d\tau} \frac{d\tau}{dl},
\]

This distribution must be corrected for any pore size dependence in the trapping probability of Ps. Thus

\[
\frac{dN_{Ps}}{dl} = \frac{dN_{void}}{dl} P_{T},
\]

where \( \frac{dN_{void}}{dl} \) is the void number-density distribution in \( l \) and \( P_{T} \) is the \( l \)-dependent specific trapping probability. If we write \( dV/dl \) as the specific void-volume distribution for the given void geometry, we have

\[
\frac{dV}{dl} = \frac{dN_{void}}{dl} V_{void} = \frac{dN_{Ps}}{dl} V_{void} P_{T}.
\]

Since Ps is formed in the solid material and must subsequently diffuse into the pores we expect Ps trapping to favor pores with large surface area. Thus, we assume \( P_{T} \propto S_{void} \) and hence we find that \( V_{void}/P_{T} \propto l \). The desired relation between specific void-volume distribution and Ps lifetime distribution for a particular void geometry is then given by

\[
\frac{dV}{dl} = \frac{dN_{void}}{dl} V_{void} \frac{dN_{Ps}}{dl} V_{void} P_{T}.
\]
assumed that Ps trapping depended on the pore surface area. Hence, the reasonable assumption that $P_T \propto S_{\text{void}}$ is important to quantitative interpretation of the pore-size distribution and future PALS work should attempt to test this hypothesis. Nonetheless, we have presented a physical model that provides the basis for transforming a Ps lifetime distribution into a pore size distribution that becomes (as it should) completely pore-model independent in the limit of large pores.

Quantitative comparison with transmission electron microscopy (TEM) presented in Ref. 4 (using a similar 10% PCL in MSSQ sample) is complicated by the fact that TEM produces a 2D projection of a 100 nm slice of the film. Further comparison of PALS results with TEM images, neutron scattering analysis, and gas absorption results is imperative for quantitative interpretation in other materials. For fully closed-pore films depth profiling with PALS can also search for depth-dependent inhomogeneities.

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